

Effect of Mg impurity on the superconductivity of metal oxides of the system Bi-Sr-Ca-Cu

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The superconducting transition temperatures T_c of the metal-oxide systems $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$ and $\text{BiSrCaMg}_x\text{Cu}_2\text{O}_y$ were found to be independent of the magnesium concentration. A strong degradation of T_c was found to occur in the system $\text{Bi}_2\text{Sr}_2\text{Ca}_{1-x}\text{Mg}_x\text{Cu}_2\text{O}_y$ as the Mg concentration was increased.

The discovery of high-temperature superconductivity in the Bi-Sr-Ca-Cu-O system,^{1,2} in which there are no rare-earth elements, has generated a considerable interest in this system. An experimental study^{1,3} of a more complex system Bi-Al-Sr-Ca-Cu-O showed that the T_c does not change appreciably as a result of the addition of aluminum.

We have synthesized a high- T_c superconducting system Bi-Sr-Mg-Ca-Cu-O and then studied it experimentally. We obtained the compositions $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$, $\text{BiSrCaMg}_x\text{Cu}_2\text{O}_y$, and $\text{Bi}_2\text{Sr}_2\text{Ca}_{1-x}\text{Mg}_x\text{Cu}_2\text{O}_y$, where $0 \leq x \leq 1.0$. The samples were synthesized by the solid-phase reaction method in air at temperatures in the range 800–870 °C. To synthesize the required compositions, we used thoroughly dried, finely divided Bi_2O_3 , MgO , and CuO oxide powders and SrCO_3 and CaCO_3 carbonate powders. The x-ray phase analysis of the synthesized powdered samples was carried out in a $\text{CuK}\alpha$ radiation using a DRON-3 diffractometer. Figure 1 shows some typical diffraction patterns of samples of various compositions.

The temperature dependence of the electrical resistance $R(T)$ was measured by the four-point method in the temperature interval 4.2–300 K at a constant current of 0.5–2 mA. Figure 2 is a plot of $R(T)/R(300 \text{ K})$ for several samples of various compositions. The temperature of the onset of the transition to the superconducting state, T_c^0 , and the temperature at which the transition ends, T_c^E , determined from these plots, are shown in Fig. 3a. As can be seen, the introduction of magnesium into the superconducting metal oxide $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$ in the amount up to $x = 0.3$ has virtually no effect on the values of T_c^0 and T_c^E . A further addition of magnesium to $x = 1.0$ changes the values of T_c^0 and T_c^E only slightly. A similar curve for the temperature of the onset of the superconducting transition and the temperature at which it ends, plotted as a function of the amount of magnesium introduced, is also appreciable to the $\text{BiSrCaMg}_x\text{Cu}_2\text{O}_y$ compositions. At 300 K the electrical resistance of the samples studied varied in the range 12–100 $\text{m}\Omega \cdot \text{cm}$.

Replacement of calcium by magnesium in the composition $\text{Bi}_2\text{Sr}_2\text{Ca}_{1-x}\text{Mg}_x\text{Cu}_2\text{O}_y$ leads to a strong degradation of T_c^0 and even stronger degra-

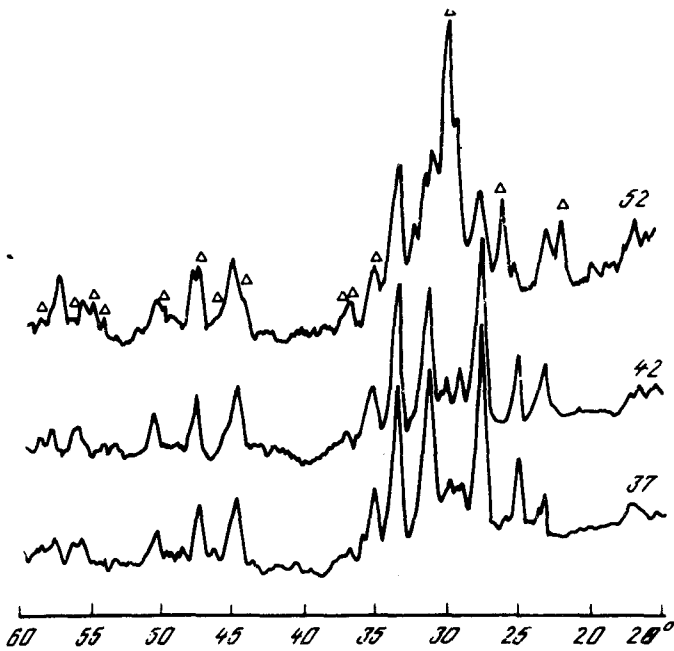


FIG. 1. Diffraction patterns of the samples. 37— $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_y$; 42— $\text{Bi}_2\text{Sr}_2\text{CaMg}_{0.5}\text{Cu}_2\text{O}_y$; 52— $\text{Bi}_2\text{Sr}_2\text{Ca}_{0.5}\text{Mg}_{0.5}\text{Cu}_2\text{O}_y$.

dation of T_0^E . After a total replacement of calcium, the composition $\text{Bi}_2\text{Sr}_2\text{MgCu}_2\text{O}_y$ remains a superconductor with $T_c^0 = 10$ K and $T_c^E = 4.2$ K.

Analysis of x-ray diffraction patterns showed that the crystal lattice of superconducting metal oxides $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$ and $\text{BiSrCaMg}_x\text{Cu}_2\text{O}_y$ has a tetragonal or pseudotetragonal structure with the parameters close to $a \approx b \approx 5.4$ Å and $c \approx 30.7$ Å, as reported in Refs. 4 and 5. The lattice parameters $a(x)$ and $c(x)$ determined by us

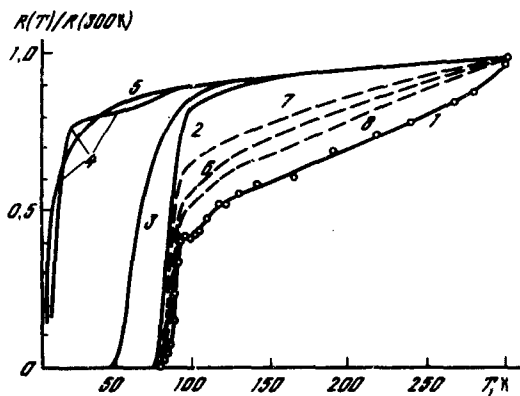


FIG. 2. $R(T)/R(300\text{K})$ curves (300 K) for samples of the system Bi-Sr-Ca-Mg-Cu-O. 1— $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_y$; 2— $\text{Bi}_2\text{Sr}_2\text{Ca}_{0.8}\text{Mg}_{0.2}\text{Cu}_2\text{O}_y$; 3— $\text{Bi}_2\text{Sr}_2\text{Ca}_{0.5}\text{Mg}_{0.5}\text{Cu}_2\text{O}_y$; 4— $\text{Bi}_2\text{Sr}_2\text{Ca}_{0.2}\text{Mg}_{0.8}\text{Cu}_2\text{O}_y$; 5— $\text{Bi}_2\text{Sr}_2\text{Mg}_x\text{Cu}_2\text{O}_y$; 6— $\text{Bi}_2\text{Sr}_2\text{CaMg}_{0.3}\text{Cu}_2\text{O}_y$; 7— $\text{Bi}_2\text{Sr}_2\text{CaMg}_{0.8}\text{Cu}_2\text{O}_y$; 8— $\text{Bi}_2\text{Sr}_2\text{CaMgCu}_2\text{O}_y$.

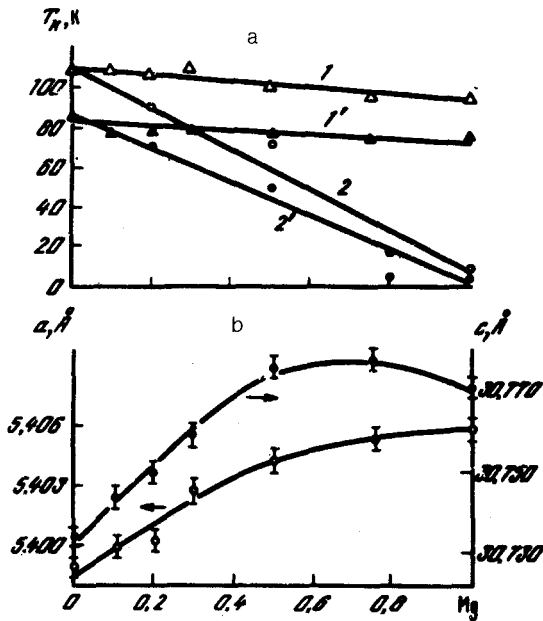


FIG. 3. Magnesium concentration versus (a) T_c^0 and T_c^E of the system $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$ (1 and 1') and of the system $\text{Bi}_2\text{Sr}_2\text{Ca}_{1-x}\text{Mg}_x\text{Cu}_2\text{O}_y$ (2 and 2'); (b) lattice constants a and c of the system $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$.

are plotted in Fig. 3b as a function of the magnesium content in the composition $\text{Bi}_2\text{Sr}_2\text{CaMg}_x\text{Cu}_2\text{O}_y$. According to our studies, it can be assumed that Mg-based phases are not formed in the metal oxide superconductors indicated above and that the lattices of the compositions which we have studied contain magnesium.

Some samples of the compositions $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_y$ and $\text{BiSrCaCu}_2\text{O}_y$ has a small amount of the high- T_c superconducting phase with $T_c^0 \approx 117$ K, as can clearly be seen on the $R(T)/R(300\text{ K})$ curves.

Study of the phase composition of samples of the system $\text{Bi}_2\text{Sr}_2\text{Ca}_{1-x}\text{Mg}_x\text{Cu}_2\text{O}_y$ showed that in addition to the formation of a high- T_c superconducting phase characteristics of $\text{Bi}_2\text{Sr}_2\text{CaCu}_2\text{O}_y$ ($x=0$), there is the formation of another phase (or phases), whose volume increases with increasing amount of magnesium.

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