

with plane-parallel end faces ( $\sim 3''$ )<sup>6</sup>. For convenience in comparing the data of Figs. 1 and 2, the numbering of the SE spectra is connected with the temperature scale of Fig. 2. The spectra 36 - 40 (Fig. 1) are shown on an enlarged scale (relative to  $\lambda$ ), so as to illustrate the switching of the SE at  $T > 600^\circ\text{K}$  to the transition I ( $11533\text{ cm}^{-1}$   ${}^4\text{F}_{3/2} \rightarrow {}^4\text{I}_{11/2}$   $2271\text{ cm}^{-1}$ ) with  $\lambda_g = 10796\text{ \AA}$ . At these temperatures,  $\lambda_g(A) = 10801\text{ \AA}$  (Fig. 2a and 2b). The flare-up of the high-temperature lines F, G, H, and I is due, on the one hand, to the change of the polarization properties of the crystal, and on the other to the temperature growth of the population of the upper level of the  ${}^4\text{F}_{3/2}$  term, and also to the resonances (for certain lines).

In conclusion, the author thanks Corresponding Member of the USSR Academy of Sciences B.K. Vainshtein and V.Ya. Khaimov-Mal'kov for a discussion of the results, G.A. Bogomolov, B.N. Grechushnikov, N.R. Ivanov, and Z.B. Perikalin for help, valuable advice, and critical remarks, and also Kh.S. Bagdasarov, under whose direction the  $\text{YAlO}_3:\text{Nd}^{3+}$  crystals were synthesized.

- [1] Kh.S. Bagdasarov and A.A. Kaminskii, ZhETF Pis. Red. 8, 501 (1969) [JETP Lett. 8, 307 (1969)].
- [2] A.A. Kaminskii and D.N. Vylegzhanin, IEEE J. Quant. Electr. QE-7, 329 (1971).
- [3] A.A. Kaminskii, Opto-electron. 3, 19 (1971).
- [4] A.A. Kaminskii, Zh. Eksp. Teor. Fiz. 58, 407 (1970) [Sov. Phys.-JETP 31, 216 (1970)].
- [5] M.J. Weber, M. Bass, K. Andringa, R.R. Monchamp, and E. Comperchio, Appl. Phys. Lett. 15, 342 (1969).

#### SUPERCONDUCTIVITY AND STRUCTURE OF COMPOUNDS BASED ON NIOBIUM AND VANADIUM

B.N. Kodess and V.Sh. Shekhtman  
 Institute of Solid State Physics, USSR Academy of Sciences  
 Submitted 4 August 1971  
 ZhETF Pis. Red. 14, No. 5, 338 - 341 (5 September 1971)

The question of the connection between the temperature  $T_c$  of the transition to the superconducting state and different structural characteristics of

<sup>6</sup>Owing to the presence of boundaries between the disoriented sections of the given sample, the He-Ne laser radiation passing through it (6328 Å) forms a system of symmetrical beams with mutually perpendicular polarization and with sharply decreasing intensity, with divergence  $\sim 1, \sim 2^\circ$ , etc., lying in one plane.

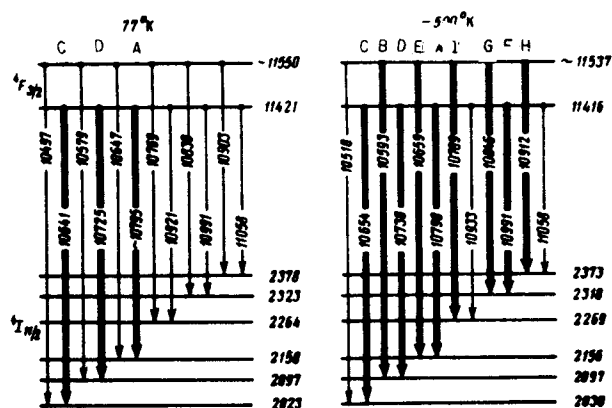


Fig. 3. Schemes of crystal splitting of the  ${}^4\text{F}_{3/2}$  and  ${}^4\text{I}_{11/2}$  terms of the  $\text{Nd}^{3+}$  ions in  $\text{YAlO}_3$  for 77 and  $\sim 500^\circ\text{K}$ . The level positions are indicated in  $\text{cm}^{-1}$ , and the transitions between them in A. The heavy arrows denote inducing transitions (scheme for  $77^\circ\text{K}$ ) registered below  $300^\circ\text{K}$  and (for  $500^\circ\text{K}$ ) above  $300^\circ\text{K}$ . The SE on line B was obtained with a laser with ( $\text{YAlO}_3:\text{Nd}^{3+} + \text{SrF}_2:\text{LaF}_2:\text{Nd}^{3+}$ ) [1].

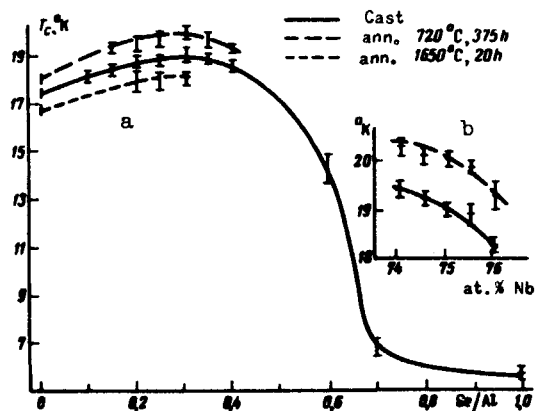


Fig. 1

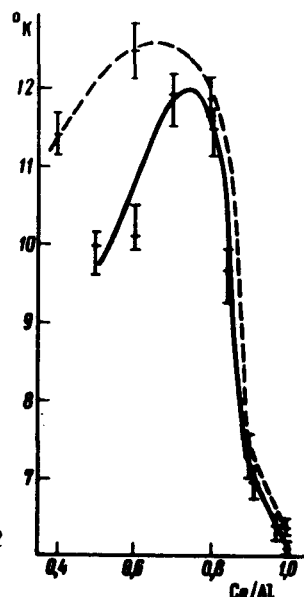


Fig. 2

Fig. 1. a - Dependence of  $T_c$  on the Al:Ge ratio, for alloys of the system Nb-Ge-Al. Nb content 75 at.%; b - dependence of  $T_c$  on the content for alloys with Al:Ge = 3:1.

Fig. 2. Dependence of  $T_c$  on the Al:Ge ratio for alloys of the V-Ge-Al system. Content 75 at. % V.

intermetallic compounds is presently extensively discussed in the literature. Compounds based on niobium and vanadium have relatively low  $T_c$ . In particular, the maximum  $T_c$  has been attained with the ternary compounds of the Nb-Ge-Al system. The reason for the appreciable rise of  $T_c$  when some of the Al atoms are replaced by Ge in the compound  $Nb_3(Ge-Al)$  are still not clear. Further experiments are needed to make it possible to assess the influence of the variation of the composition of the phases, the structural transitions, ordering phenomena, etc. on the parameters of the superconducting transitions. In the present investigation we studied a number of alloys containing compounds with structure A-15 in the ternary systems niobium-germanium-aluminum and vanadium-germanium-aluminum.

The procedures used to prepare the samples and to measure their properties were described earlier [1].

Figure 1 shows the data obtained on the concentration dependence of  $T_c$  in the investigated alloys. The highest transition temperature ( $20.3^\circ K$ ) was obtained at Al:Ge = 3:1 for an annealed alloy containing 74 at. % Nb (Fig. 1b). It is interesting that a slight decrease of the Nb content at this ratio leads to a rise of  $T_c$ , whereas according to the data of [2] an increase of  $T_c$  corresponds to the region of excess of Nb relative to the  $Nb_3(Ge-Al)$  composition. A composition change corresponding to isomorphic substitution of Al(Ge) leads to the appearance of a maximum on the concentration dependence of  $T_c$  (Figs. 1a and 2). We note that to attain a maximum for the V-Ge-Al system it is necessary to have the inverse ratio, Al:Ge = 1:3 (Fig. 2).

The noticeable dependence of  $T_c$  on the ratio of the aluminum to germanium and the rise of the transition point on approaching the value 3:1 were

previously observed for Nb-Ge-Al alloys in [2]. On the basis of these data, the authors of [2, 3] made the natural assumption that the ordering of the Ge and Al atoms possibly exerts an influence, but no definite experimental confirmation of this assumption was obtained. It was therefore of interest to carry out a neutron-diffraction analysis of the compound  $V_3(\text{Ge-Al})$ , bearing in mind that the vanadium atoms have a small scattering amplitude and the diffraction pattern should be quite sensitive to the distribution of the germanium and aluminum. Neutron-diffraction patterns were obtained for a sample with composition  $V_3(\text{Ge}_{0.75}\text{Al}_{0.25})$  in the cast state and after annealing at  $850^\circ$  for 375 hours. The line intensity was compared with the calculation. A typical section of the neutron-diffraction pattern is shown in Fig. 3. The low intensity of the (210) line indicates that the V atoms are almost entirely in the position 6(c). The absence of superstructure lines (see the positions (311), (222), (331) marked by the superior arrows) indicates that there is no ordering in accordance with the  $\text{Fe}_3\text{Al}$  scheme in the Ge and Al sublattice. Experiments were also performed for the purpose of establishing the connection between the characteristics of the annealing and the transition temperature. According to the data shown in Figs. 1 and 2, annealing at a given temperature and time exerts a similar influence on  $T_c$  in the entire concentration interval: after low-temperature annealing,  $T_c$  rises noticeably, and after high-temperature annealing a certain lowering of  $T_c$  takes place in comparison with the same samples in the cast state. We note that the ordering of the Al and Ge atoms should have led to a change of  $T_c$  in a definite interval of compositions.

Preliminary data on the influence of the annealing time were obtained for the alloy with composition  $\text{Nb}_3(\text{Al}_{0.75}\text{Ge}_{0.25})$  (see the table). We see that a rise in the transition temperature corresponds to annealing at  $950^\circ$  for not more than 90 min, after which the effect of the annealing decreases appreciably, as was reported also for an annealing temperature  $1000^\circ$  in [4]. The debyegramms have shown that an annealing time that leads to a rise in the transition temperature is accompanied in many cases by smearing of the lines. The results do not permit us as yet to state with assurance that the initial stages of the decay (of the aging type) influence the transition temperature, but they indicate that further research in this direction is desirable.

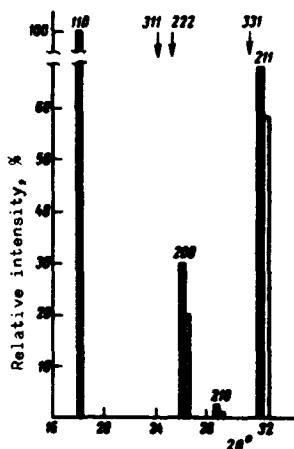


Fig. 3. Calculated (light bars) and experimental (dark bars) intensities of the alloy  $V_3(\text{Al}_{0.25}\text{Ge}_{0.75})$  on the neutron-diffraction pattern.

It was also of interest to ascertain the existence of a martensitic transformation in the compounds of the investigated ternary systems. To this end, low-temperature x-ray investigations were made of the alloy  $\text{Nb}_3(\text{Ge}_{0.3}\text{Al}_{0.7})$ . At  $20^\circ\text{K}$ , we observed a change in the form of

Dependence of the change of  $T_c$  on the annealing time at  $950^\circ\text{C}$

$t$ , min	$\Delta T_c$ , $^\circ\text{K}$
3	0.2
6	0.3
30	0.6
90	0.3
180	0.2
300	0.2

the diffraction lines, making it possible to assume, by analogy with [5 - 7], the existence of a structure transition with tetragonal strain  $\epsilon \approx -(3.1 \pm 1.2) \times 10^{-3}$ . We note that the value of  $\epsilon$  lies between the values  $\epsilon$  for  $V_3Si$  and  $Nb_3Sn$ . The sign of  $\epsilon$  coincides with that for  $Nb_3Sn$  [7]. It is interesting that the nuclear-resonance investigations at 24°K clearly point to a change in the symmetry of the lattice in the same samples, which also indicates the presence of a transition [8] not observed in  $Nb_3Al$  [9]. This circumstance confirms the important role of correlation between the superconducting and martensitic transitions in high-temperature superconductors of the A-15 type.

The authors are grateful to A.G. Rabinkin for affording the possibility of measuring  $T_c$ , and to R.A. Sizov for recording the neutron diffraction patterns.

- [1] B.W. Kodess, Phys. Stat. Sol. (a) 4, K109 (1971).
- [2] G. Arrhenius, E. Corenzwitt, R. Fitzgerald, G.W. Hull, H.L. Luo, B.T. Matthias, and W.H. Zachariasen, Proc. Nath. Acad. Sci. 61, 621 (1968).
- [3] N.E. Alekseevskii, N.V. Ageev, and V.F. Shamrai, Izv. AN SSSR, Neorg. mat. 12, 2156 (1966).
- [4] I. Rozoca, Zs. Physik. 237, 432 (1970).
- [5] B.W. Batterman and C.S. Barret, Phys. Rev. 137, 3447 (1966).
- [6] L.J. Viland, R.N. Cohen, and W. Rehlwald, Phys. Rev. Lett. 26, 7, 373 (1971).
- [7] H.W. King, F.H. Cocks, and J.T.A. Pollock, Phys. Lett. A26, 2, 77 (1967).
- [8] B.N. Kodess, V.B. Kurizin, and B.N. Tretjakov, Phys. Lett. 1 (1971) (in print).
- [9] R.H. Willens, T.H. Geballe, A.C. Gossard, J.P. Maita, and A. Menth, Solid State Comm. 7, 837 (1969).

#### CRITICAL TEMPERATURE OF AMORPHOUS SUPERCONDUCTING FILMS

I.O. Kulik

Physico-technical Institute of Low Temperatures, Ukrainian Academy of Sciences

Submitted 5 July 1971

ZhETF Pis. Red. 14, No. 5, 341 - 344 (5 September 1971)

Naugle and Glover [1] observed a decrease of the critical temperature of the superconducting transitions of amorphous (i.e., having very high resistance) films of Bi and Ga, condensed on a low-temperature substrate, in inverse proportion to their thickness  $d$ . This means that  $\Delta T_c$  is proportional to the value of the normal resistance. In the present paper we propose an explanation for this effect, based on the assumption that the lowering of  $T_c$  is due to fluctuations of the electromagnetic field in the film.

We define the point of transition into the superconducting phase as the point where the ordering parameter  $\psi$  exhibits perturbations that increase with time. According to the nonstationary Ginzburg-Landau equation [2] we have

$$\frac{\partial \psi}{\partial t} + \Gamma \psi - D \left( \nabla - \frac{2ie}{c} \mathbf{A} \right)^2 \psi = 0, \quad (1)$$

where

$$\Gamma = \frac{8}{\pi} (T - T_{c0}), \quad D = \frac{1}{3} v_F \ell,$$

and  $\ell$  is the mean free path. At  $\vec{A} = 0$ , the transition point  $T = T_{c0}$  is determined from the condition  $\Gamma = 0$ . When the fluctuation fields ( $\vec{A}$ ) are taken into