High-temperature solid/melt nanocomposites

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The interfacial energy of nanomaterials can be decreased by grain boundary wetting. Most effective is grain boundary wetting by chemically compatible melt in the two-phase area of the phase diagram where the solid and melt are in equilibrium. The possibility of the thermodynamically stable solid/melt nanocomposite existence in the two-phase area of the phase diagrams where the solid and melt are in equilibrium, is shown.

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Nanomaterials [1] have a high interfacial energy and, therefore, they are thermodynamically unstable and experience recrystallization problem at $0.2\text{--}0.3\,T_m$ (T_m is the melting temperature). The interfacial energy can be decreased sufficiently by grain boundary (GB) wetting. Most effective is GB wetting by chemically compatible melt in the two-phase area of the phase diagram where the solid and melt are in equilibrium [2]. In this case, the solid/melt interfacial energy can be decreased largely ($\sim 1\text{--}80\,\text{mJ/m}^2$) [3-4].

Here we show the possibility of the thermodynamically stable solid/melt nanocomposite existence in the two-phase area of the phase diagrams where the solid and melt are in equilibrium. The calculated, within framework of a low-angle GB wetting model [5], values of the solid/melt interfacial energies are typical for liophilic colloids [3]. This opens the way to the design of the thermodynamically stable high-temperature nanomaterials.

GB wetting by chemically compatible melt in the two-phase area of the phase diagrams has been established for both ceramic and metallic materials [6, 7]. Thermodynamic condition of the GB wetting can be written as [8]:

$$\gamma_{ab} > 2\gamma_{sl},\tag{1}$$

where γ_{gb} and γ_{sl} are the specific free energies at the grain boundary and solid/melt interface, respectively. The γ_{gb} depends on the misorientation angle, θ . Lowand high-angle GBs are distinguished by the θ . Lowangle GBs refer to low-energy; high-angle to high-energy (with the exception of the so-called special GBs). As GBs have a wide spectrum of energies, the temperature of GB wetting will differ: the lower γ_{gb} , the higher temperature. Because size of the blocks (subgrains) in solids is $10\text{-}100\,\mathrm{nm}$, let us consider the thermodynamic opportunity of low-angle GB wetting. Read and Shockley [9]

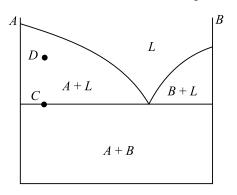
have proposed a formula for γ_{gb} calculation as a function of the θ :

$$\gamma_{gb} = \frac{Ga}{4\pi(1-\sigma)}\theta(\beta - \ln \theta), \tag{2}$$

where G is the rigidity modulus, a is the lattice constant, σ is the Poisson's ratio, and β is approximately 0.25. According to the equation (2), the γ_{gb} increases with increasing θ and reaches a maximum, $\gamma_{gb\,(\text{max})}$. If the θ corresponding to the $\gamma_{gb\,(\text{max})}$ marks as θ_{max} , than the eq. (2) can be transformed to [10]:

$$\frac{\gamma_{gb}}{\gamma_{gb\,(\text{max})}} = \frac{\theta}{\theta_{\text{max}}} \left(1 - \ln \frac{\theta}{\theta_{\text{max}}} \right) \tag{3}$$

Estimation shows that the γ_{gb} values for low-angle GBs, $10'' \leq \theta \leq 1^{\circ}$, is varied from $1 \, \mathrm{mJ/m^2}$ to $150 \, \mathrm{mJ/m^2}$ at $\theta_{\mathrm{max}} = 30^{\circ}$ and $\gamma_{gb \, (\mathrm{max})} = 10^3 \, \mathrm{mJ/m^2}$. Consequently, the thermodynamic condition of low-angle GB wetting (1) is fullfield at $0.5 \, \mathrm{mJ/m^2} \leq \gamma_{sl} \leq 75 \, \mathrm{mJ/m^2}$. These γ_{sl} values on order of magnitude are typical for liophilic colloids [3]. In particular, the liophilic colloid may form in the two-phase area of the phase diagram (Figure) where the solid A or B and melt are in equilibrium. The



Phase diagram

formation of the high-temperature liophilic colloid could be explained as follows. It is known that the GB wetting by eutectic melt occurs in the two-phase area of

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the phase diagrams (point C in Figure) [2]. With further rise in the temperature, γ_{sl} is decreased. If the γ_{sl} reaches a value of $0.5\,\mathrm{mJ/m^2}-75\,\mathrm{mJ/m^2}$, the lowangle GB wetting by chemically compatible melt occurs (for example, point D in Figure). At that, the partial dissolution of the solid grains A takes place. As the result, the thermodynamically stable high-temperature solid/melt nanocomposite may exist.

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